# **Experimental Characterization and Modeling of Twin-Screw Extruder Elements for Pharmaceutical Hot Melt Extrusion**

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In this study we characterized various screw elements of a co-rotating twin-screw extruder used for pharmaceutical hot melt extrusion (HME) and measured the pressure characteristic, i.e., the correlation between the axial pressure gradient and the material throughput in a completely filled screw section at different screw speeds. A typical HME matrix material, Soluplus, was used for the experiments and its required rheological properties were determined. A three-parameter model based on a dimensionless formulation of the measured quantities was used. These parameters could not be determined uniquely by fitting to experimental data. Therefore we developed an approach to approximate one empirical parameter based on the mechanistic consideration of a pressure-driven channel flow. The model was extended to account for the variable melt temperature. The results confirmed the expected tendencies and established an essential input parameter set for one-dimensional simulations of co-rotating twin-screw extruders. © 2013 American Institute of Chemical Engineers AIChE J, 59: 4440–4450, 2013

Keywords: hot melt extrusion, screw element, pressure characteristic, rheology, soluplus

#### Introduction

Hot melt extrusion (HME) has been used for many decades in various industries (e.g., plastics and food) and, in recent years, has become an important pharmaceutical manufacturing operation.<sup>1</sup> Extrusion is defined as the process of forcing a material through a die under controlled conditions and high pressure. HME applies temperature levels above the melting (or softening) point in order to extrude the material in a viscous state.<sup>2</sup> In pharmaceutical manufacturing, HME can potentially increase the bioavailability of poorly soluble drugs,<sup>3</sup> form solid solutions and amorphous solid dispersions<sup>4</sup> and enhance the product quality due to the mixing efficiency of extruders.<sup>2</sup> Moreover, in a single step raw materials can be converted into a final delivery form, for example, through calandering.

Screw extruders (single-, twin-, and multi-screw) are the most common extrusion devices, which combine various process steps such as mixing, melting, conveying, degassing, and dissolution of particles in a single device. These properties increase the manufacturing efficiency and reduce operation costs. In pharmaceutical manufacturing, typically co-rotating

twin-screw extruders with self-cleaning screws and modular screw design are used, that is, the screws are divided into different types of screw elements (e.g., conveying elements and kneading elements), which can be individually arranged to accommodate actual process requirements. While this feature offers high flexibility, achieving optimal process conditions is a challenge as developing optimal screw configurations requires experience and/or experimental work based on trial and error.

Mechanistic models can provide useful insights into the effects of various screw configurations, improve understanding of the underlying mechanisms, and reduce the experimental and empirical effort. Although different models describing the flow field in screw extruders have been developed, not all aspects of the extrusion process can be predicted in detail. For example, to date no detailed methods exist for the simulation of the flow in partially filled screws. The involved physical phenomena (non-Newtonian flow, wall slip, small gaps, free surfaces, etc.) are highly complex and first-principle design methods are not available.<sup>5</sup>

Existing mechanistic models of the entire HME process are based on one-dimensional (1-D) models of the flow along corotating twin screws, resolving the pressure and filling ratio along the screw axis.<sup>6-9</sup> These models basically account for the pressure-throughput correlation of single screw elements, describing the pressure gradient vs. flow rate in completely filled screw sections.<sup>5,10-14</sup> The resulting predictions of the material transport along the screw have been used to develop

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models of the residence time distribution in extruders.<sup>15–19</sup> In recent years, this approach has been extended to models of reactive extrusion.<sup>20–22</sup> Spatially resolved simulations of the flow field in completely filled screw sections have been performed using computational fluid dynamics (CFD),<sup>14,23–28</sup> but they cannot be applied to an entire twin screw due to the required calculation effort and the challenge of partially filled screw sections in three-dimensional (3-D) simulations. Thus, simplified descriptions of extrusion processes are still prevalent.

Pawlowski<sup>10</sup> showed that for single screw extruders and Newtonian fluids in the creeping flow regime, the axial pressure gradient is linearly correlated with the throughput, and described the correlation by an empirical model termed "pressure characteristic", based on dimensionless numbers. As the pressure characteristic is determined by the detailed screw geometry, the used empirical parameters are called "screw parameters". Kohlgrüber and Wiedmann<sup>5</sup> applied this concept to co-rotating twin-screw extruders and extended it to shear-thinning polymer melts described by the Carreau model.

In our work, we developed a 1-D simulation for co-rotating twin-screw extruders based on the pressure-characteristic model by Kohlgrüber and Wiedmann<sup>5</sup>. To simulate a real extrusion process, screw parameters for all used screw elements must be provided as input parameters. Thus, we present pressure-characteristic measurements for different screw elements and the resulting screw parameters using Soluplus, a typical matrix material for pharmaceutical HME. The required temperature-dependent rheological material parameters of Soluplus have been measured. In the extrusion experiments, we found a slight variation of the melt temperature with the throughput, which is not negligible due to the strong temperature dependency of the viscosity. To account for this effect, the pressure characteristic was corrected for the temperaturedependence of the material parameters. Moreover, we developed an approach to estimate the empirical shear-rate parameter  $A_3$  to overcome an underdetermined fitting problem. In summary, input parameters for the 1-D simulation of corotating twin-screw extruders were established.

## **Mathematical Model**

#### Pressure-characteristic model

The description of Pawlowski<sup>10</sup> and Kohlgrüber and Wiedmann<sup>5</sup> is based on a dimensionless formulation of the volumetric throughput and the axial pressure gradient in completely filled screw sections (similarly to Ref. 11), where the dimensionless throughput  $\dot{V}^*$  is defined as the volumetric throughput  $\dot{V}$  over the screw speed n and the nominal screw diameter D cubed

$$\dot{V}^* = \frac{\dot{V}}{nD^3}.\tag{1}$$

For measurement purposes, the axial pressure gradient  $\partial p/\partial x$  is approximated by the quotient of pressure difference and axial distance  $\Delta p/\Delta x$  (see Figure 7). The dimensionless axial pressure gradient  $\Delta p^*/\Delta x^*$  is defined using the dynamic viscosity of the material  $\eta$ , the screw speed n, and the nominal screw diameter D as follows

$$\left(\frac{\Delta p^*}{\Delta x^*}\right) = \frac{\Delta p}{\eta \cdot n} \cdot \frac{D}{\Delta x} \tag{2}$$

The correlation between dimensionless throughput  $\dot{V}^*$  and dimensionless axial pressure gradient  $(\Delta p^*/\Delta x^*)$  is termed

pressure characteristic. Using the described dimensionless quantities, Pawlowski<sup>10</sup> experimentally showed that the pressure characteristic of a certain screw element is invariant for different screw speeds n, screw diameters D, and viscosities  $\eta$  under the following conditions:

- Creeping flow, that is, sufficiently low Reynolds number  $Re = \rho \cdot n \cdot D^2 / \eta$  ( $\rho$  being the density). The limiting Reynolds number depends on the screw geometry. In a specific case (D = 60 mm), Re < 100 satisfied this condition. Typically, high-viscosity materials used in HME generally lead to Reynolds numbers below 1 and, thus in technically relevant cases the creeping flow regime can be assumed.
  - Newtonian fluid
- Geometrical similarity of the entire flow-relevant geometry.

Furthermore, it was demonstrated that for Newtonian fluids in the creeping flow regime, the pressure characteristic follows a linear function, which is described as<sup>5,10</sup>

$$\frac{\Delta p^*}{\Delta x^*} = \frac{\Delta p}{\eta \cdot n} \cdot \frac{D}{\Delta x} = A_2 \cdot \left(1 - \frac{\dot{V}^*}{A_1}\right) \tag{3}$$

where  $A_1$  and  $A_2$  are the screw parameters, that is, the axis intercepts of the linear function (i.e.,  $\Delta p^*/\Delta x^* = 0$  for  $\dot{V}^* = A_1$  and  $\Delta p^*/\Delta x^* = A_2$  for  $\dot{V}^* = 0$ ). Thus, parameter  $A_1$  represents the throughput in a completely filled screw section without back pressure (i.e., inherent conveying capacity), whereas  $A_2$  represents the pressure gradient in a completely filled screw section conveying toward a closed die. An approximation for  $A_1$  is given as<sup>29</sup>

$$A_1 \approx \frac{1}{2} \frac{A_{\rm cr} \cdot T_{\rm S}}{D^3} \tag{4}$$

where  $A_{\rm cr}$  is a free cross-section area illustrated as a shaded area in Figure 2 and  $T_{\rm S}$  is the screw pitch. In reality, typical materials used for HME are not Newtonian fluids but rather shear-thinning materials that can often be described via the Carreau model for constant temperature<sup>5,30</sup>

$$\eta(\dot{\gamma}) = \frac{\eta_0}{\left[1 + \frac{|\dot{\gamma}|}{\dot{\gamma}_{\text{crit}}}\right]^m} \tag{5}$$

where  $\eta_0$  is the zero-shear-rate viscosity,  $\dot{\gamma}_{\rm crit}$  is the critical shear rate and m the Carreau exponent. Kohlgrüber and Wiedmann<sup>5</sup> extended the linear pressure characteristic for Newtonian fluids to non-Newtonian fluids described by the Carreau model. For this purpose, the Newtonian viscosity in Eq. 3 was substituted by Eq. 5

$$\frac{\Delta p^*}{\Delta x^*} = \frac{\Delta p}{\left[1 + \frac{|\dot{y}|}{r_{\text{crit}}}\right]^m \cdot n} \cdot \frac{D}{\Delta x} = A_2 \cdot \left(1 - \frac{\dot{V}^*}{A_1}\right) \tag{6}$$

Then the dimensionless axial pressure gradient was redefined for a Carreau fluid (indicated by index C) using the zero-shear-rate viscosity  $\eta_0$  as

$$\frac{\Delta p_{\text{C}^*}}{\Delta + x^*} = \frac{\Delta p}{\eta_0 \cdot n} \cdot \frac{D}{\Delta x} = \frac{A_2 \cdot \left(1 - \frac{\dot{V}^*}{A_1}\right)}{\left[1 + \frac{|\dot{\gamma}|}{\dot{\gamma}_{\text{crit}}}\right]^m} \tag{7}$$

This pressure-characteristic function is nonlinear and predicts lower pressure gradients compared to the linear function in the Newtonian case caused by the shear-thinning effect of

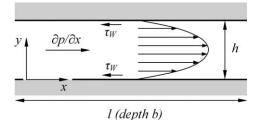


Figure 1. Two-plate model.

the melt. However, the issue of how to determine the shear rate  $\dot{\gamma}$  is critical for a correct representation of the problem. In the real flow field around the screws, the shear rate varies strongly, for example, in the smallest gaps of  $O(10^{-4}-10^{-3})$  m), higher shear rates can be expected than in the screw channels of  $O(10^{-3}-10^{-2})$  m). Thus, using a single value for the shear rate in Eq. 7 ("representative shear rate"  $\dot{\gamma}_r$ ) is a simplification. However, Kohlgrüber and Wiedmann<sup>5</sup> created a model that describes the representative shear rate proportional to the pressure-induced backflow  $(1-\dot{V}^*/A_1)$ . At the inherent conveying point  $(\dot{V}^*=A_1)$ , where no pressure-induced backflow occurs, the representative shear rate is zero. The representative shear rate increases with the increasing backflow. Furthermore, the shear rate is set to be proportional to the screw speed and an empirical fitting parameter  $A_3$ 

$$\dot{\gamma}_{\rm r} = A_3 \cdot n \cdot \left(1 - \frac{\dot{V}^*}{A_1}\right) \tag{8}$$

Substituting this in Eq. 7 leads to the following pressurecharacteristic function<sup>5</sup>

$$\frac{\Delta p_{\text{C}^*}}{\Delta x^*} = \frac{\Delta p}{\eta_0 \cdot n} \cdot \frac{D}{\Delta x} = \frac{A_2 \cdot \left(1 - \frac{\dot{V}^*}{A_1}\right)}{\left[1 + \frac{A_3 \cdot n}{\dot{\gamma}_{\text{crit}}} \cdot \left|1 - \frac{\dot{V}^*}{A_1}\right|\right]^m} \tag{9}$$

Equation 9 can be used to fit the measured pressure gradients for shear-thinning fluids described by the Carreau model<sup>5</sup>. Kohlgrüber and Wiedmann<sup>5</sup> also presented a formulation equivalent to Eq. 9, which is based on the representative viscosity  $\eta_r$  (with  $\eta_r = \eta(\dot{\gamma}_r)$  from Eq. 5 and  $\dot{\gamma}_r$  from Eq. 8)

$$\frac{\Delta p^*}{\Delta x^*} = \frac{\Delta p}{\eta_{\rm r} \cdot n} \cdot \frac{D}{\Delta x} = A_2 \cdot \left(1 - \frac{\dot{V}^*}{A_1}\right) \tag{10}$$

Here, the shear-thinning effect is incorporated into the representative viscosity  $\eta_r$ , leading to a representation in which the measured data points of shear-thinning fluids are located along the linear pressure-characteristic function of Newtonian fluids.

#### Approximation of A<sub>3</sub>

Our study showed that establishing the three screw parameters  $(A_1, A_2, A_3)$  by fitting the measured data leads to an underdetermined problem, that is, different combinations of  $A_1$ ,  $A_2$ , and  $A_3$  can describe the measurements equally well. Furthermore, the calculated pressure gradient is significantly less sensitive to variations of parameter  $A_3$  than of parameters  $A_1$  and  $A_2$  (which can easily be demonstrated by varying parameters in Eq. 9). Thus, we approximate the value of  $A_3$  based on a simplified model of a two-dimensional (2-D) channel flow between two plates (see Figure 1). Assuming a stationary and developed flow, the force balance relates the pressure gradient  $\partial p/\partial x$  and the wall shear stress  $\tau_{\rm W}$  as

$$\frac{\partial p}{\partial x} \cdot l \cdot b \cdot h = 2 \cdot \tau_{W} \cdot b \cdot l \tag{11}$$

The wall shear stress can be written as product of wall shear rate and the corresponding viscosity  $\tau_W = \eta_W \cdot \dot{\gamma}_W$  with  $\eta_W = \eta(\dot{\gamma}_W)$ . If so, Eq. 11 leads to

$$\dot{\gamma}_{W} = \frac{1}{\eta_{W}} \frac{\partial p}{\partial x} \cdot \frac{h}{2} \tag{12}$$

Considering Eq. 8 and substituting the term in brackets with Eq. 10 results in

$$\dot{\gamma}_{\rm r} = \frac{1}{\eta_{\rm r}} \cdot \frac{\Delta p}{\Delta x} \cdot \frac{A_3}{A_2} \cdot D \tag{13}$$

Comparing the simplified two-plate model with the real situation of a twin-screw, and assuming, that  $\dot{\gamma}_{\rm W}$  and  $\eta_{\rm W}$  are estimates for  $\dot{\gamma}_{\rm r}$  and  $\eta_{\rm r}$ , Eqs. 12 and 13 yield the following approximation for  $A_3$ 

$$A_3 \approx \frac{A_2 \cdot h}{2 \cdot D} \tag{14}$$

In this equation, the gap distance h is not known. In fact, the gap distance between the screw and the barrel in an extruder varies depending on the cross-sectional profile of the screws (see Figure 2). To create an average, the parameter h was estimated as ratio between the gap cross-section area  $A_{\rm cr}$  (i.e., shaded area in Figure 2 top) and the gap length  $l_{\rm gap}$  (estimated as the entire length of the lines shown in Figure 2 bottom). Parameter h was calculated as

$$h = \frac{A_{\rm cr}}{l_{\rm gap}} \tag{15}$$

The use of Eqs. 14 and 15 reduces the number of free parameters to two  $(A_1 \text{ and } A_2)$ , and the solution of the fitting problem is unique.

#### Temperature dependency

As expected, the melt temperature was not constant during our measurements. In addition to the expected dependence of the melt temperature on the screw speed, the melt temperature also slightly varied with the throughput. Due to the strong temperature dependency of the viscosity, we could

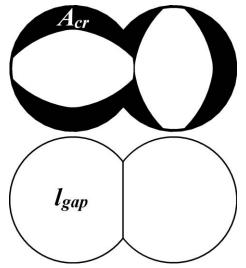


Figure 2. Schematic of gap cross-section area  $A_{cr}$  (top) and gap length  $I_{gap}$  (bottom).

**Table 1. Menges Model Parameters** 

	Solid	Liquid
$K_1$ [bar cm <sup>3</sup> /g]	29716	29519
$K_2$ [bar cm <sup>3</sup> /g°C]	0.35526	1.1618
$K_3$ [bar]	1671.5	2614.4
$K_4$ [bar]	34326	34783
	Transition	temperature
$K_8$ [°C]	72.729	-
$K_9$ [°C/bar]	0.020912	

not neglect this effect and incorporated the temperature dependency in the pressure characteristic of Kohlgrüber and Wiedmann, <sup>5</sup> using a temperature shift factor  $a_{\rm T}$  in the Carreau model Eq.  $5^{31}$ 

$$\eta(\dot{\gamma}, T) = \frac{\eta_0 \cdot a_{\rm T}}{\left(1 + \frac{|\dot{\gamma}| \cdot a_{\rm T}}{\dot{\gamma}_{\rm crit}}\right)^m} \tag{16}$$

The temperature shift factor in our case can be described by the Williams–Landel–Ferry (WLF) equation (often used for amorphous thermoplasts<sup>30</sup>), where  $C_1$  and  $C_2$  are material constants, that is,

$$a_{\mathrm{T}} = \exp\left[\frac{-C_1 \cdot (T - T_{\mathrm{r}})}{C_2 + T - T_{\mathrm{r}}}\right] \tag{17}$$

Using this and the approximation of  $A_3$  (Eq. 14), the pressure-characteristic function Eq. 9 becomes

$$\frac{\Delta p_{\text{C}^*}}{\Delta x^*} = \frac{\Delta p}{\eta_0 \cdot a_{\text{T}} \cdot n} \cdot \frac{D}{\Delta x} = \frac{A_2 \cdot \left(1 - \frac{\dot{V}^*}{A_1}\right)}{\left[1 + \frac{A_2 \cdot n \cdot a_T}{\dot{\gamma}_{\text{crit}}} \cdot \frac{h}{2D} \cdot \left|1 - \frac{\dot{V}^*}{A_1}\right|\right]^m}$$
(18)

Eq. 10 remains unchanged. The representative viscosity is described by Eq. 16 for nonisothermal conditions.

## Nonconveying elements

The proposed concept was shown by Kohlgrüber and Wiedmann<sup>5</sup> for actively conveying elements. Here, we derived a modification for nonconveying elements (e.g., kneading elements with 90° offset angle) for which parameters  $A_1$  and  $A_2$  are zero (i.e., the inherent conveying capacity and the ability to generate pressure are zero). In the limit of  $A_1$  and  $A_2$  going to zero in Eq. 18, the following equation is recovered, where  $A_0 = A_2/A_1$  for  $A_1$ ,  $A_2 \rightarrow 0$ 

$$\frac{\Delta p_{\text{C}^*}}{\Delta x^*} = \frac{\Delta p}{\eta_0 \cdot a_{\text{T}} \cdot n} \cdot \frac{D}{\Delta x} = \frac{-A_0 \cdot \dot{V}^*}{\left[1 + \frac{A_0 \cdot n \cdot a_{\text{T}}}{\dot{\gamma}_{\text{crit}}} \cdot \frac{h}{2D} \cdot |\dot{V}^*|\right]^m}$$
(19)

Similarly, Eq. 8, including the approximation of  $A_3$  (Eq. 14), and Eq. 10 can be used for nonconveying screw elements

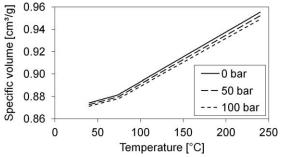


Figure 3. Specific volume as a function of temperature and pressure.

$$\dot{\gamma}_{\rm r} = A_0 \cdot n \cdot \frac{h}{2D} \cdot \dot{V}^* \tag{20}$$

$$\frac{\Delta p^*}{\Delta x^*} = \frac{\Delta p}{\eta_{\rm r} \cdot n} \cdot \frac{D}{\Delta x} = -A_0 \cdot \dot{V}^* \tag{21}$$

where the representative viscosity  $\eta_r = \eta(\dot{\gamma}_r, T)$  is described by Eq. 16.

#### **Materials**

In our experiments, the pharmaceutical-grade matrix material Soluplus was used, which is a polyvinyl caprolactam-polyvinyl acetate-polyethylene glycol graft copolymer specifically designed for HME. Soluble in water and many organic solvents, it is well-suited as a solubilizer for poorly soluble drugs. More detailed information can be obtained from the manufacturer. To evaluate our measurements, the equation of state and viscosity depending on the temperature and shear rate are required.

#### Equation of state

The equation of state of Soluplus was measured with a PVT-device of type PVT100 (SWO Polymertechnik GmbH, Germany) according to the standard ISO 17744.<sup>33</sup> Isobarcooling with 6 K/min at five different pressure levels from 200 to 1200 bar was used. The measured data were fitted with the Menges model<sup>34</sup> (parameters  $K_1$ – $K_4$  are different for solid and liquid state), which describes the specific volume v [m³/kg] depending on pressure p and temperature T

$$v = \frac{K_1}{p + K_4} + \frac{K_2 \cdot T}{p + K_3}.\tag{22}$$

The transition temperature between solid and liquid states (in this case the glass transition temperature) depending on pressure p is described by

$$T_{\text{trans}} = K_8 + K_9 \cdot p \tag{23}$$

The parameters established through the experiments are provided in Table 1. The resulting dependence of the specific volume on temperature and pressure is shown in Figure 3. Clearly, the pressure impact is low (variation < 1% between 0 and 100 bar) compared to the temperature variability (about 10% between 40 and 240°C). A discontinuity in the curves indicates the glass transition point of Soluplus (approximately  $70^{\circ}\text{C}^{32}$ ).

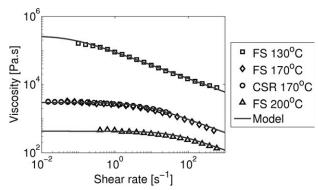


Figure 4. Viscosity over shear rate.

The shown measurements are frequency sweeps (FS) and a rotation test under controlled shear rate (CSR).

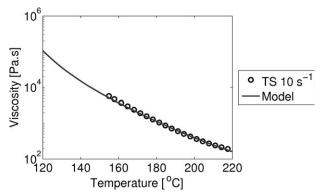


Figure 5. Viscosity over temperature.

The shown measurement is a temperature sweep (TS).

#### Melt viscosity

The viscosity of the Soluplus melt was measured with a MCR301 (Anton Paar, Austria) equipped with electric temperature control systems P-EDT400 and H-EDT400 and a cone-to-plate measurement body with a diameter of 25 mm and 1° cone angle. Different types of measurements were performed: three frequency sweeps (FS) at 130, 170, and 200°C, a rotation test with controlled shear rate (CSR) at 170°C, and a temperature sweep (TS) at 10 rad/s. Although the shear viscosity (i.e., the result of CSR tests) was required, FS tests are preferable as they allow a wider measurement range of angular frequency (and shear rate). As shown in Figure 4, FS and CSR at 170°C are in agreement, confirming the validity of the Cox-Merz rule<sup>31</sup>, under which the result of the FS (complex viscosity depending on the angular frequency) is equal to the CSR result (shear viscosity depending on shear rate). In addition, to confirm the temperature dependency established via the FS at three different settings, a TS at a constant angular frequency of 10 rad/s was performed (Figure 5).

The model (Carreau and WLF equation, Eqs. 16 and 17) was fitted, and the yielding parameters are shown in Table 2. Calculated model curves vs. the measured data are shown in Figures 4 and 5. The measurements were in agreement with the data supplied by the manufacturer.<sup>32</sup>

# **Screw Characterization Experiments**

Experiments were performed to measure the axial pressure gradient and material temperature for different throughputs, screw speeds, and screw element configurations. The extruder used was a ZSK 18 from Coperion (Germany) with a nominal screw diameter D=18 mm. The characterization was performed for six different types of screw elements (all two-flighted), that is, two conveying screw elements termed "24/24" and "16/16" with different pitches of 24 and 16 mm, respectively, two kneading elements "KB 45/5/8" and "KB 45/5/8 LH" (45° offset angle, 5 discs, 8 mm length, right- and left-handed), a kneading element "KB 90/5/8" and

**Table 2. Viscosity Parameters** 

$\eta_0$	2999.9 Pa·s
γ̈ <sub>crit</sub>	$5.7852 \text{ s}^{-1}$
m	0.39489
$T_{\rm r}$	170°C
$C_1$	10.7203
$C_2$	135.4020°C

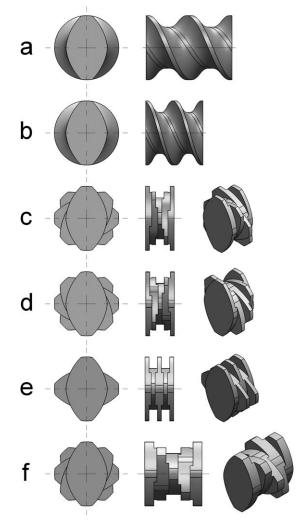


Figure 6. ZSK18 screw elements evaluated in our study.

(a) 24/24, (b) 16/16, (c) KB 45/5/8, (d) KB 45/5/8 LH, (e) KB 90/5/8, and (f) KB 45/5/16.

a kneading element "KB 45/5/16". All elements are depicted in Figure 6. From the geometry of the cross section (Figure 2), a free cross-section area  $A_{\rm cr} = 187.8~{\rm mm}^2$  was calculated. The gap length  $l_{\rm gap}$  was 101.6 mm, leading to an average gap size  $h = 1.847~{\rm mm}$  based on Eq. 15.

Two pressure sensors (melt pressure sensors from Gefran, Italy; I-Series, 0–35bar) were installed close to the screw end at an axial distance of  $\Delta x = 43.2$  mm (see Figure 7). The built-in temperature sensors of the extruder are planar

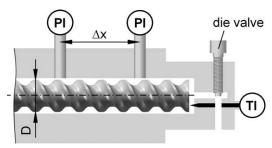


Figure 7. Experimental set-up.

PI is the pressure indicator and TI the temperature indicator.

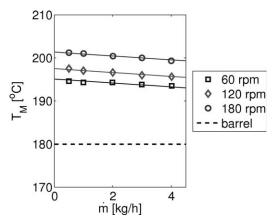


Figure 8. Melt temperature (points: measurement, lines: fit) over throughput at different screw speeds for the conveying screw element 24/24.

with the wall (similar to the pressure sensors) and their readings are strongly influenced by the barrel temperature. These sensors always measure some average of melt and barrel temperature. Thus, another temperature sensor (fast-response thermometer Thermapen from ETI Electronic Temperature Instruments, UK) was placed in the clearance volume directly behind the screw, which was the only option for inserting it directly in the melt flow. To avoid wall contact, the shaft of this additional temperature sensor was covered with a polytetrafluoroethylene (PTFE) casing. The measured temperature was significantly higher (approximately 3–10°C) than that indicated by the built-in temperature sensors, obviously significant viscous dissipation occurs. The screw section between both pressure sensors was configured with the screw element type being investigated. To ensure that the filled screw section extended over both pressure sensors, an adjustable die valve was constructed (Figure 7).

The barrel temperature was set to 180°C. Vacuum devaporization was used (200 mbar) to avoid foaming caused by moisture in Soluplus. The measurements were performed for three screw speeds (60, 120, and 180 rpm) and throughputs of 0.5–4 kg/h and 0.5–2 kg/h for the conveying and kneading elements, respectively. Each change of throughput or screw speed was maintained for at least 15 min to achieve a steady state before collecting data for 6 min using the SIPAT System (Siemens, Belgium) with a data acquisition rate of 5 s. As the measured pressure data showed significant oscillations, steady state in this case meant that the pressure difference was oscillating around a constant value over the considered time period. The measured temperature did not show oscillations and was recorded manually by the operator.

## **Results and Discussion**

# Measured data

The resulting data of the measured melt temperature and pressure gradient for the 24/24 screw element (Figure 6a) are shown in Figures 8 and 9. The melt temperature  $T_{\rm M}$  was significantly higher than the barrel temperature due to the considerable amount of heat generated by viscous dissipation, even in the case of conveying elements. As expected, the temperature increased with the increasing screw speed due to greater dissipation and slightly decreased with increasing throughput due to the reduced axial pressure gra-

dient (i.e., reduced viscous dissipation) and an increased cooling due to convective heat transport. The axial pressure gradient  $\Delta p/\Delta x$  was averaged over 6 min for each measured data point, the error bars indicate the standard deviation of the observed oscillations. With increasing screw speed, the axial pressure gradient increased due to increased shear forces balanced by pressure gradients and decreased with increasing throughput due to a decreased backflow in the completely filled section, as also shown by the linear pressure-characteristic model for Newtonian fluids (Eq. 3). The dependency of the axial pressure gradient on the throughput becomes weaker with increasing screw speed. At 180 rpm, the pressure gradient was rather constant for the 24/24 screw element. This can be explained by the viscosity increase due to the corresponding temperature decrease of approximately 2°C, which dominates the slight decrease of the axial pressure gradient with increasing throughput expected due to the pressure characteristic (Note that also at 180 rpm, an inherent conveying capacity exists, i.e., a throughput, where the axial pressure gradient becomes zero. However, that throughput is relatively high at 180 rpm).

The data for all other screw elements were qualitatively similar and are not shown in detail. Specifically, we did not find a significant difference in the melt temperature for all investigated screw elements, indicating, that the heat generation by viscous dissipation is not significantly different, even for kneading elements. Note, that the local viscous dissipation rate is proportional to viscosity and shear rate squared  $(q_{\rm Diss} = \eta \cdot \dot{\gamma}^2)$ . The dominating shear rates are mainly caused by the screw rotation and the gap distance between screws and barrel, which is equal for all investigated screw elements due to the invariant cross-section profile of the screws. In the case of kneading elements, the offset of the single discs causes additional gaps in the intermeshing region of the screws where high shear rates occur. However, this does not dominate the heat generation, as only a small amount of material is located in the crossover of offset regions and the intermeshing region.

#### Fitted model

The measurements were used to fit parameters  $A_1$  and  $A_2$  in Eq. 18 for all of the investigated screw elements and parameter  $A_0$  in Eq. 19 for the nonconveying screw element KB 90/5/8. The specific volume per Eq. 22, which was

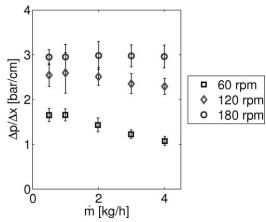


Figure 9. Axial pressure gradient over throughput at different screw speeds for the conveying screw element 24/24.

Table 3. Fitted Screw Parameters and Relative Variations Due to  $h \pm 50\%$ 

Screw Element	$A_1$	$A_2$	$A_3$
24/24	0.3593 +15.3% / -7.6%	766.5 ±18.8%	39.33 +77% / -59%
16/16	0.2257 + 6.4% / -3.4%	$808.6 \pm 18.9\%$	41.49 +77% / -59%
KB 45/5/8	0.1545 + 7.4% / -5.0%	$259.1 \pm 10.2\%$	13.29 +65% / -55%
KB 45/5/16	0.1448 + 4.8% / -3.6%	$217.7 \pm 8.5\%$	11.17 +63% / -54%
KB 90/5/8	$A_0 = 1214.4 \pm 2.9\%$		

required to convert mass throughput  $\dot{m}$  [kg/h] into volumetric throughput  $\dot{V}$  [m³/h], was evaluated at the measured melt temperature  $T_{\rm M}$  and the pressure p=0, that is, the (small) pressure dependency was neglected (see Figure 3). The material parameters given in Tables 1 and 2 were used, and the temperature shift factor  $a_{\rm T}$  per Eq. 17 was calculated based on the measured melt temperature  $T_{\rm M}$ .

The fitted parameters for all of the investigated screw elements are shown in Table 3. In order to demonstrate the sensitivity of the fitted screw parameters  $A_0$ ,  $A_1$ , and  $A_2$  on the developed estimate of parameter h and  $A_3$ , the estimate h was varied by  $\pm 50\%$  (1.847 mm  $\pm 0.924$  mm). The resulting deviations in  $A_0$ ,  $A_1$ ,  $A_2$ , and  $A_3$  are shown by relative errors in Table 3. The values of parameter  $A_3$  deviated in the range of 54–77%, while the corresponding deviations of  $A_0$ ,  $A_1$ , and  $A_2$ were in the range of 3, 3–15, and 8–19%, respectively. This indicates that deviations in  $A_0$ ,  $A_1$ , and  $A_2$ , which were caused by deviations in  $A_3$ , were significantly lower than deviations in  $A_3$ , that is, the model is hardly sensitive to the estimated parameter  $A_3$ . As such, the used approximation is a valid approach and it can be expected that the data reported here allow the simulation of extruders and the prediction of their performance within technical accuracy ranges.

Using the determined parameters  $A_0$ ,  $A_1$ , and  $A_2$ , the pressure-characteristic curves were calculated according to Eq. 18 for conveying elements and Eq. 19 for nonconveying element KB 90/5/8. The calculated curves describe changes in the dimensionless axial pressure gradient as a function of throughput and screw speed. The corresponding melt temperature is required for the evaluation of Eqs. 18 and 19 due to the temperature dependency of the viscosity, but cannot be predicted by the pressure-characteristic model. Thus, the melt temperature was determined from a linear fit of the measured temperature data  $T_{\rm M}(n,\dot{m}) = a_{\rm n} - b \cdot \dot{m}$  (see solid lines in Figure 8, fitted parameters  $a_{\rm n}$  and b are provided in Table 4).

A comparison between the calculated pressure-characteristic curves and the measured data is shown in Figures 10–15 for all investigated screw elements. The dashed lines represent the linear model of Newtonian fluids with axis intercepts  $A_1$  and  $A_2$  (calculated in Eq. 3). The discrepancy between that and the nonlinear curves fitting the measured data represents the effect of shear-thinning and temperature variations.

Clearly, the model fits the measured data well and is a good basis for extruder design and performance prediction. The model fit is better for the conveying screw elements 24/

**Table 4. Fitted Temperature Parameters** 

	$a_n$ [°C]			
Screw element	60 rpm	120 rpm	180 rpm	b [°C.h/kg]
24/24	195.05	197.48	201.33	0.4545
16/16	191.73	195.88	200.48	0.3566
KB 45/5/8	194.02	198.33	200.97	0.3360
KB 45/5/16	193.91	199.97	202.74	1.0076
KB 90/5/8	195.97	199.90	203.53	0.7424

24 (Figure 10) and 16/16 (Figure 11) than for the kneading elements (Figures 12-15). This can be explained by the stronger relative pressure oscillations observed for the kneading elements (represented by error bars). The ratio of the inherent conveying capacity (i.e., parameter  $A_1$ ) for both of the investigated conveying screw elements 0.3593/ 0.2257 = 1.59 is comparable with the ratio of their pitch 24/ 16 = 1.5. This is in agreement with the data reported by Kohlgrüber et al.<sup>29</sup>, who stated that  $A_1$  is approximately proportional to the screw pitch. Both  $A_1$  parameters are comparable to the approximation calculated in Eq. 4 (0.3864 and 0.2576). In contrast,  $A_2$  of the 24/24 and 16/16 are relatively similar, that is, the generated pressure gradient at zero throughput is not significantly different. Kohlgrüber and Wiedmann<sup>5</sup> demonstrated that  $A_2$  generally increases with the decreasing pitch, except for cases when there is a large gap between the screw and the barrel (typical for relatively small machines, such as ZSK 18 that we used).

Parameters  $A_1$  and  $A_2$  are significantly lower for the kneading elements KB 45/5/8 (Figure 12) and KB 45/5/16 (Figure 13) than for the conveying elements due to the geometry of the kneading elements (Figure 6) with discontinuous steps rather than a smooth surface that reduces the conveying effect and pressure build-up. KB 45/5/16 has a slightly decreased  $A_1$  and  $A_2$  compared to KB 45/5/8, that is, the stretched geometry of KB 45/5/16 resulted in reduced conveying and pressure build-up.

The measured data of the screw element KB 45/5/8 LH were not fitted, as they should be identical to the pressure characteristic of KB 45/5/8 except for the inversed direction (negative throughput) due to its backward conveying effect (also see the corresponding literature<sup>5,11</sup>). The data points for KB 45/5/8 LH and KB 45/5/8 are shown together in Figure 14. Although the data approximately match at zero throughput, the axial pressure gradient of KB 45/5/8 LH showed an unexpected discrepancy compared to the fitted curves of KB

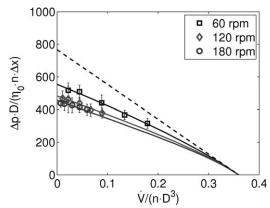


Figure 10. Pressure characteristic based on  $\eta_0$  for the conveying screw element 24/24.

Dots: measurements, solid lines: model for Soluplus, dashed line: model for Newtonian fluids.

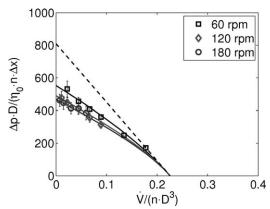


Figure 11. Pressure characteristic based on  $\eta_0$  for the conveying screw element 16/16.

Dots: measurements, solid lines: model for Soluplus, dashed line: model for Newtonian fluids.

45/5/8, even a decreasing axial pressure gradient with increasing negative throughput in some regions.

As expected, the kneading element KB 90/5/8 (Figure 15) had a pressure-characteristic line passing the origin as its geometry does not cause a preferred conveying direction and is thus nonconveying. The curve slope was similar to the other kneading elements, meaning that the change in the pressure drop with varying throughput was similar.

The resulting linear pressure-characteristic curves of all of the investigated screw elements are compared in Figure 16. For a clear graphical representation, the dimensionless axial pressure gradient  $\Delta p^*/\Delta x^* = \Delta p \cdot D/(\eta_{\rm f} \cdot n \cdot \Delta x)$  according to Eq. 10 (Eq. 21 for nonconveying elements) was used, in contrast to  $\Delta p_{\rm C}^*/\Delta x^* = \Delta p \cdot D/(\eta_0 \cdot n \cdot \Delta x)$  as used for Figures 10–15. The obvious difference is the shear-rate-dependent representative viscosity  $\eta_{\rm r}$  (in contrast to  $\eta_0$ ), which leads to a linear correlation, as the shear-thinning effect of the melt rheology is incorporated in  $\eta_{\rm r}$ . Thus, all data points of a screw element are located along the pressure characteristics for Newtonian fluids in Figure 16 which is independent of the screw speed.

Although the relative oscillations of the kneading elements were significantly higher than those of the conveying elements, the absolute oscillations of the dimensionless pressure gradient

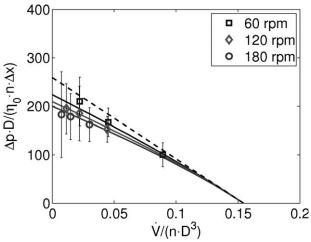


Figure 12. Pressure characteristic based on  $\eta_0$  for the kneading element KB 45/5/8.

Dots: measurements, solid lines: model for Soluplus, dashed line: model for Newtonian fluids.

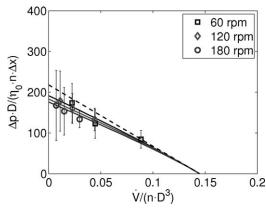


Figure 13. Pressure characteristic based on  $\eta_0$  for the kneading element KB 45/5/16.

Dots: measurements, solid lines: model for Soluplus, dashed line: model for Newtonian fluids.

were similar for the investigated screw elements (see Figure 16). The cross-section profile of the screw elements was equal for all considered elements. This indicates that the observed oscillations of the axial pressure gradient are caused by the rotation of the screws, which led to pressure maxima in front of and minima behind the screw flights with respect to the rotational direction (this conjecture, however, cannot be proven based on the shown data, as the data acquisition rate of 5 s does not resolve the circulation time of the screws). However, this effect has been shown by 3-D simulations of twin screws<sup>5</sup> for the pressure distribution over the cross section.

## **Summary and Conclusions**

This work presents the first detailed experimental characterization of completely filled screw elements for pharmaceutical HME with Soluplus on a co-rotating twin-screw extruder. Specifically, the pressure-characteristic (i.e., the correlation between the axial pressure gradient and the volumetric throughput in a completely filled screw section) was measured at different screw speeds and for different screw elements. The required rheological properties and the specific volume of Soluplus were determined in detail the first time by precise

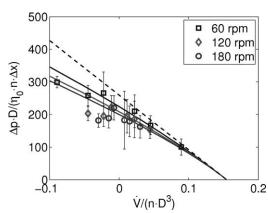


Figure 14. Pressure characteristic based on  $\eta_0$  for the kneading element KB 45/5/8 together with KB 45/5/8 LH.

Dots: measurements, solid lines: model for Soluplus, dashed line: model for Newtonian fluids.

measurements and described via the Carreau model combined with the WLF equation, and the Menges model, respectively. The pressure-characteristic model  $^{5,10}$  was fitted to describe the measured data. As fitting of three parameters was found to be an underdetermined problem, an approximation of the parameter  $A_3$ , which has a lower impact on the results than parameters  $A_1$  and  $A_2$ , was developed based on a mechanistic understanding of the pressure-driven flow. The effect of the approximation was significantly lower for  $A_1$  and  $A_2$  than for  $A_3$ , that is, deviations in  $A_1$  and  $A_2$  were lower than deviation in  $A_3$  introduced by the approximation. As it was established that the melt temperature was not constant for variable throughput and screw speed, a temperature dependency was included into the model using a linear fit of the measured temperature over the throughput.

The results show significantly different pressurecharacteristic curves and corresponding screw parameters for the investigated screw elements. A comparison of the screw parameters for the different elements confirm the expected tendencies, such as the proportionality of the inherent conveying capacity and the screw pitch of conveying screw elements, as well as the same axial pressure gradient for rightand left-handed elements of the same type (i.e., inverse conveying direction) at zero throughput.

More detailed studies (e.g., via numerical simulations) are required to investigate why the backward-conveying kneading element KB 45/5/8 LH deviated from the model curves of element KB 45/5/8 and, in some regions, had decreased pressure gradients at increased throughputs. Extensive 3-D simulations (e.g., CFD) of the investigated screw elements are necessary to confirm the determined screw parameters and pressure-characteristic curves and to analyze the developed approximation of  $A_3$ . This is currently the aim of our work.

Although we used only one type of material and one type of extruder for our measurements, the applicability of the results to other cases is supported by the following reasons. The dimensionless formulation of the model leads to scale invariance, that is, geometrically similar screw elements at different length scales have identical screw parameters. Moreover, the screw parameters are invariant of the melt viscosity in the creeping flow regime (which is typically the case for the highly viscous materials used in HME). Both statements were experimentally verified for single-screw

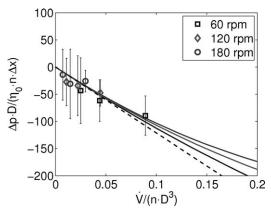


Figure 15. Pressure characteristic based on  $\eta_0$  for the kneading element KB 90/5/8.

Dots: measurements, solid lines: model for Soluplus, dashed line: model for Newtonian fluids.

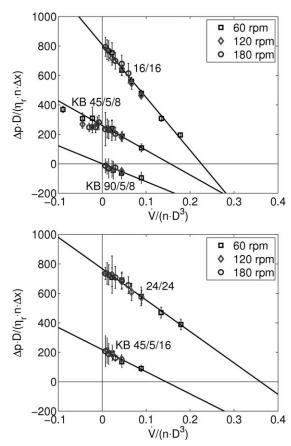


Figure 16. Pressure-characteristic curves based on  $\eta_r$  for all of the investigated screw elements in comparison (dots: measurements, lines: model).

extruders by Pawlowski<sup>10</sup>. Consequently, the reported screw parameters can be applied to different screws, if they are geometrically similar. The used pressure-characteristic model was developed from the linear model for Newtonian fluids. Incorporating the melt rheology (the Carreau model in our case), it can be used for non-Newtonian fluids. However, the screw parameters  $A_1$  and  $A_2$  always characterize the axis intercepts of the linear model for Newtonian fluids. Thus,  $A_1$  and  $A_2$  are independent of the used material. This was not shown in our present work. Numerical studies with different rheologies could verify this assumption and open a way of determining such parameters for other types of screw elements without performing experiments.

The determined parameters confirm an essential premise for a 1-D model of a co-rotating twin-screw extruder, which cannot predict the transport processes caused by the screw elements based directly on the 3-D geometry. Using the obtained screw parameters, such a model can predict process variables (e.g., filling ratio, pressure and melt temperature along the twin screw). Although the shown experiments have exclusively been performed for a completely filled screw section, the results can be used to estimate filling ratio and pressure in partially filled screw sections. Axial pressure gradients in partially filled screw sections are zero due to the connected gas phase. The dependency of the filling ratio on the throughput can be estimated linearly filling filling filling filling filling ratio = 1 at

the inherent conveying capacity. Moreover, based on the calculated flow rates along the screw such a 1-D model can predict residence time distributions, that is, characterize the mixing performance of the process.

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#### **Notation**

## Latin symbols

 $A_0$  = screw parameter for nonconveying elements, - $A_1, A_2$  = screw parameters for conveying elements, - $A_3$  = empirical shear rate parameter, -

 $a_{\rm T}$  = temperature shift factor (Carreau model, WLF equation), -

 $C_1$ ,  $C_2$  = parameters of the WLF equation D = nominal screw diameter, m

h = gap distance, m

 $K_1$ ,  $K_2$ ,  $K_3$ ,  $K_4$ ,  $K_8$ ,  $K_9$  = parameters of the Menges model

 $\dot{m}$  = mass throughput [kg/s]

m =Carreau exponent, -

n =screw speed, s

p= pressure, Pa  $\frac{\Delta p^*}{\Delta x^*}=$  dimensionless axial pressure gradient based on the representative viscosity  $\eta_r$  or Newtonian viscosity  $\eta$ , -

 $\frac{\Delta p_{C^*}}{\Delta r^*}$  = dimensionless axial pressure gradient based on the zero-shear rate viscosity  $\eta_0$  (Carreau model). -

Re =Reynolds number, -

 $T = \text{temperature}, ^{\circ}\text{C}$ 

 $T_{\rm M}$  = measured melt temperature,  $^{\circ}{\rm C}$ 

 $T_{\rm r}$  = reference temperature (WLF equation), °C

 $T_{\rm S}$  = screw pitch, m

 $T_{\text{trans.}}$  = transition temperature (Menges model), °C V = volumetric throughput, m<sup>3</sup>/s

 $\dot{V}^*$  = dimensionless volumetric throughput, -

 $v = \text{specific volume, m}^3/\text{kg}$ 

x = axial coordinate, m

#### Greek letters

 $\dot{\gamma}$  = shear rate, s<sup>-1</sup>

 $\dot{\gamma}_{crit}$  = critical shear rate (Carreau model), s<sup>-1</sup>

 $\dot{\gamma}_r$  = representative shear rate, s

 $\eta$  = dynamic viscosity, Pas

 $\eta_0$  = zero-shear-rate viscosity (Carreau model), Pas

 $\eta_{\rm r}$  = representative viscosity, Pas

 $\rho$  = density, kg/m<sup>3</sup>

 $\tau_{\rm W}$  = wall shear stress, Pa

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